

Structural, electronic, cohesive, and elastic properties of diamondlike allotropes of crystalline C₄₀Andrey N. Enyashin^{1,2,*} and Alexander L. Ivanovskii²¹*Physikalische Chemie, Technische Universität Dresden, D-01062 Dresden, Germany*²*Institute of Solid State Chemistry, Ural Branch of the Russian Academy of Sciences, 620041 Ekaterinburg, Russia*

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The atomic models of diamondlike allotropes of crystalline C₄₀ (hyperdiamond, hyperlonsdaleite and so-called autointercalated hyperdiamond) are offered and using the density-functional based tight-binding method their structural, cohesive, elastic, and electronic properties are predicted as well as their theoretical x-ray diffraction spectra are derived. Among these allotropes the autointercalated hyperdiamond C₄₀ has a unique combined type of both covalent and van der Waals bonding and shows the highest density and hardness.

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Since the discovery of fullerene C₆₀ (Ref. 1) and the synthesis of this molecule and some other stable fullerenes C_n ($n > 60$) in macroscopic quantities,^{2,3} intensive effort has been devoted to the research of their solid derivatives. Among them to the most known systems belong the pure or doped C₆₀ molecular crystals—fullerites. The lattices of these crystals are composed of the fullerenes, which have closed electronic shells and interact by means of weak van der Waals forces.^{4,5}

An alternative group of fullerene-based solids is presented by crystals with strong covalent interactions between fullerene cages. The formation of these crystals (so-called covalently bonded fullerites) is a characteristic feature⁶ of small fullerenes C_{n<60} with open electronic shells.⁷ Among these crystals the C₂₈-based structures with diamondlike lattices, which have been called as hyperdiamonds, have attracted a significant attention. Numerous theoretical investigations on fullerene C₂₈ hyperdiamonds and related systems (for example, endohedral metallofullerenes Ti@C₂₈, Zn@C₂₈, or Na@C₂₈H₄ as building blocks of fullerites) have been reported recently, and a set of their interesting electronic, elastic, and cohesive properties was predicted.⁷⁻¹⁵ Moreover, the low mass density and large internal surface suggest future applications of these systems as catalysts, nanosieves, or gas storage materials.¹⁵

The possibility of C₂₈ to form diamondlike lattices is based on the unique structural and electronic properties of C₂₈ cage with tetrahedral symmetry T_d .^{6,7} This open-shell molecule contains four unpaired electrons localized at four pentagon triplets in a tetrahedral arrangement. This feature makes the high reactive T_d -C₂₈ fullerene similar to a sp^3 -hybridized carbon “superatom,” which is able to form chemical bonds by linking with other four molecules C₂₈ and build diamondlike lattices.

Among other small fullerenes (C_{n<60}) T_d -symmetrical C₄₀ cage is similar to T_d -C₂₈, because it has four unpaired electrons in an 5A_2 open-shell ground state. Thus, C₄₀ behaves as a reactive hollow superatom with effective valence of 4.^{6,7,16-18} The cage of T_d -C₄₀ has 12 C₅ pentagons and 10 C₆ hexagons. There are three types of nonequivalent groups of carbon atoms: C(1), C(2), and C(3). From them four C(1) atoms are placed at the vertices of a tetrahedron, which are the triplets of C₅ pentagons, as depicted in Fig. 1. Such a structure possesses a significant potential for bonding with four atoms or molecules outside of the C₄₀ cage and forma-

tion of thermodynamically stable systems (in particular, diamondlike solids), because of the presence of four dangling bonds located at its tetrahedron apices.

In this Brief Report the atomic models of three diamondlike allotropes of crystalline C₄₀ are offered and their structural, cohesive, elastic, and electronic properties are predicted by means of the density-functional based tight-binding (DFTB) theory. Two of them are analogical to crystalline C₂₈ allotropes⁷⁻¹⁵—hyperdiamond C₄₀ (HD) and hyperlonsdaleite C₄₀ (HL), and are classified by the space groups $Fd\bar{3}m$ and $P63/mmc$, respectively. These allotropes contain voluminous cavities between covalently bonded cages C₄₀, Fig. 1. This feature is attractive for an intercalation of these materials by various atoms and molecules. Moreover, the simple estimations show that the radii of these cavities within lattices (~ 5.4 nm) are higher than the radii of C₄₀ (~ 3.3 nm) forming the lattices, i.e., can be easy of access for other C₄₀ fullerenes. In other words, diamondlike C₄₀ phases are capable for autointercalation.

Thus, as a third crystalline C₄₀ allotrope we suggest so-called autointercalated hyperdiamond (AHD), whose lattice may be described as C₄₀ HD, where the cavities are com-

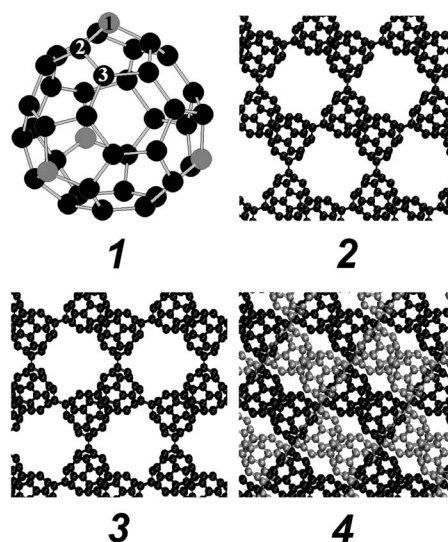


FIG. 1. Fullerene C₄₀ (1) and the fragments of the optimized lattices for crystalline C₄₀ allotropes: hyperdiamond (2), hyperlonsdaleite (3), and autointercalated hyperdiamond (4). For C₄₀ cage three nonequivalent types of carbon atoms C(1–3) are marked.

TABLE I. Formation energies (E_{form} , in kJ/mol), lattice constants (a and c , in nm), mass density (ρ , in g/cm³), bulk moduli (B , in GPa), and band gap (E_g , in eV) of the crystalline C₄₀ allotropes according to DFTB calculations.

System	$-E_{\text{form}}$	a	c	ρ	B	E_g
HD C ₄₀	427.04	1.8924		0.94	29.22	1.01
(HD C ₂₈) ^a	(602.80)	(1.5952)		(1.10)	(45.54)	(1.80)
HL C ₄₀	422.83	1.3386	2.1860	0.78	30.25	1.01
(HL C ₂₈)	(598.70)	(1.1286)	(1.8429)	(1.10)	(45.90)	(1.76)
AHD C ₄₀	427.59	1.8922		1.88	60.31	0.94

^aIn parentheses—the data for crystalline C₂₈ allotropes as obtained at the same DFTB level. (Refs. 14 and 15).

pletely occupied with the same fullerenes C₄₀, Fig. 1. Also, the structure of C₄₀ AHD can be imagined as two sublattices of C₄₀ HD, which are inserted each inside other. Unit cells of HD and AHD include two and four fullerenes C₄₀, i.e., contain 80 and 160 carbon atoms, respectively.

Our calculations were performed within the DFTB,¹⁹ which was used earlier for the simulations of diamondlike allotropes of crystalline C₂₈.^{14,15} The dispersion corrected modification of DFTB with the implementation of an empirical dispersion term parametrized in the framework of the universal force field (UFF) method was used for correct description of weak van der Waals interactions.²⁰ This method has been found to describe the structure and energy of carbon nanostructures in good agreement with experiment and higher-level theoretical methods.

The calculated properties of the above mentioned three C₄₀ allotropes: formation energies (E_{form}), lattice constants (a and c , which was kept constant and equal $\sqrt{8/3}a$ as for the ideal lonsdaleitlike lattice), mass density (ρ), and bulk moduli (B) are listed in Table I in comparison with C₂₈ HA and HL as obtained at the same DFTB level.¹⁴

To estimate the relative stability of C₄₀ allotropes, their Gibbs free energies ($G = \Delta H + P \cdot V - T \cdot S$) should be calculated. Since our band structure calculations are performed at zero temperature and zero pressure, G becomes equal to the enthalpy of formation, ΔH , which can be estimated from the total energies of C₄₀ allotropes and its constituents—free fullerenes C₄₀. The formation energies (E_{form}) of C₄₀ allotropes with respect to the “free” fullerenes C₄₀ are defined as $E_{\text{form}} = \{E_{\text{tot}}(\text{C}_{40}^{\text{cryst}}) - NE_{\text{tot}}(\text{C}_{40}^{\text{free}})\} / N$, where E_{tot} are the total energies of a unit cell of C₄₀ allotropes and free fullerene C₄₀ at their optimized geometries. In this way, negative value E_{form} indicates that it is energetically favorable for given reagents to mix and form the stable crystals, and *vice versa*. The results (Table I) show that E_{form} for all crystalline C₄₀ phases are negative, i.e., their formation is favorable. Among them the high-dense allotrope AHD is the most stable. However, the difference in the formation energies between AHD and HD is small: $\Delta E_{\text{form}} \sim 0.55$ kJ/mol of C₄₀ molecules. It can be explained considering the nature of interaction between C₄₀ cages in these crystals. In C₄₀ HD and HL, like in their topological analog—C₂₈ HD and HL—the neighboring cages form the covalentlike C(1)–C(1) bonds. For autointer-

calated hyperdiamond C₄₀ a rough estimation gives that the distance between two C₄₀ cages belonging to different diamondlike sublattices is equal to 0.4 nm. This value is comparable with van der Waals gap, i.e., with the interlayer distance in graphite (or interlayer distance in coaxial multiwalled nanotubes or multishell fullerenes), when between two graphene layers appears van der Waals bonding.^{1–4} Thus, besides strong covalent C(1)–C(1) bonds AHD phase is characterized by weak van der Waals interaction between the fullerenes of different sublattices.

Therefore, C₄₀ AHD may be considered as a type of fullerite, where the nature of interaction between molecules combines both types of bonding, which are presented in molecular crystals (C_n≥60-based fullerites) and in alternative group of C_n<60-based covalently bonded fullerites.

Let us discuss the predicted properties of C₄₀ allotropes. Bulk moduli B of C₄₀ HD and HL became smaller than for topologically identical C₂₈ HD and HL, and are twice less than B for C₄₀ AHD. This is a consequence of the much lower density of chemical bonds in C₄₀ HD and HL compared with C₂₈ HD and HL and C₄₀ AHD. Note that the bulk moduli of both isostructural C₄₀ and C₂₈ allotropes HD and HL are smaller than the bulk modulus of C₄₀ AHD. This sequence correlates with change of their mass density ρ , see Table I.

The calculated band structures of the crystalline C₄₀ allotropes are shown in Fig. 2, and their total and partial densities of states (DOSs) are in Fig. 3. All crystals C₄₀ are semiconducting materials with very similar band gaps of about 1.0 eV, see Table I. Besides, crystalline C₄₀ HD and HL with higher amount of sp^2 hybridization of atoms and larger part of π component in the bonding have smaller gaps (about 0.75 eV less) than their crystalline C₂₈ analog. This result agrees well with the known tendency¹⁴ of fullerite gap reduction by decrease of the content of sp^3 covalent bonds between C_n molecules and the growth of a system of benzene rings.

The occupied bands in the range from -9 to -2 eV consist of the C 2s and C 2p states responsible for σ and π bonds within and σ bonds between the C₄₀ cages. The shape of DOS for C₄₀ fullerites is characterized by numerous well-defined peaks below Fermi level, which are associated with π bonds of network of aromatic hexagons within C₄₀ cages. The distributions of the C 2s and C 2p densities of states (Fig. 3) confirm predominantly covalent bonding character in C₄₀ allotropes. The difference in the shape of DOS between various solid C₄₀ allotropes does not have a pronounced character because of weak van der Waals interaction between the walls of C₄₀ cages, which is not able to perturb the electronic structure considerably.

The above presented results demonstrate the similarity in the electronic structure and stability of C₄₀ allotropes proposed. To clarify the possibility of an experimental identification of these allotropes, which could be synthesized, for example, by deposition of C₄₀ fullerenes on some surfaces, we have simulated their theoretical x-ray diffraction (XRD) patterns using the optimized geometries. Assuming the polycrystallinity of a material, the average scattering power I , in electron units, is given by the Debye formula

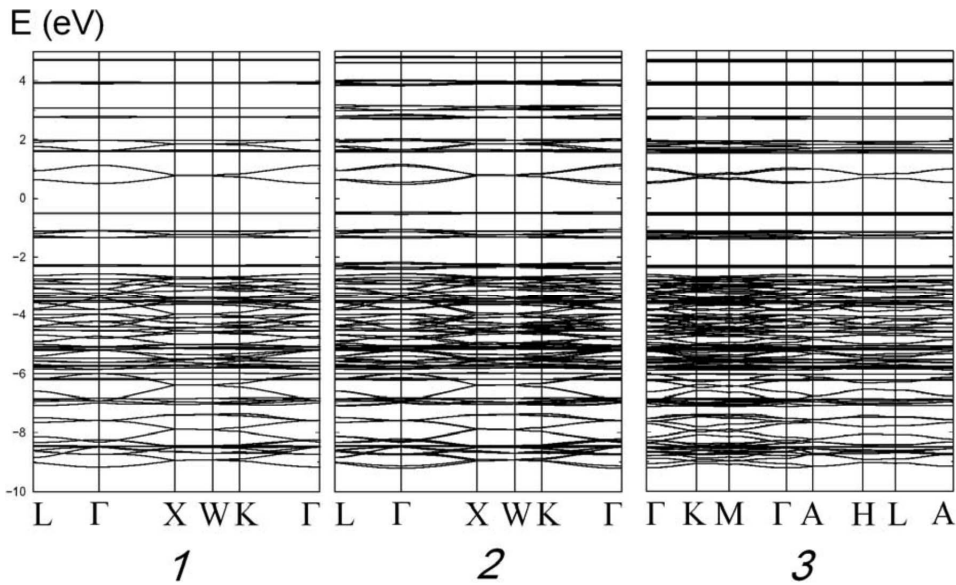


FIG. 2. Band structures for crystalline C_{40} allotropes: hyperdiamond (1), autointercalated hyperdiamond (2), and hyperlonsdaleite (3). The energy is given relative to the Fermi energy $E_F = 0$ eV.

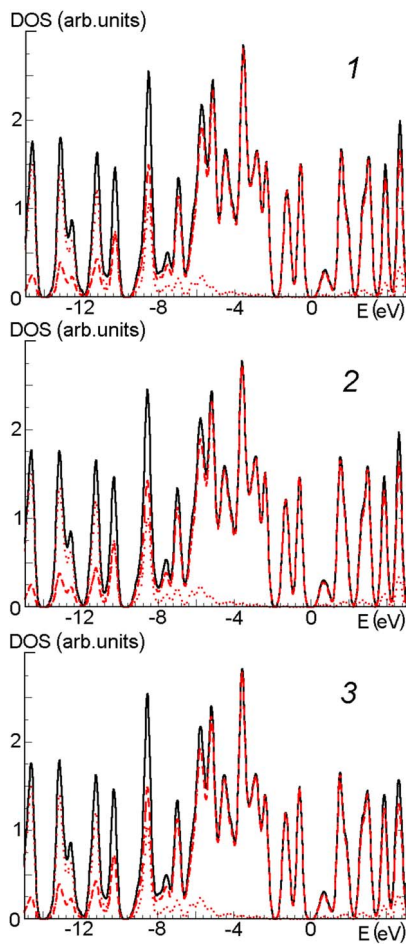


FIG. 3. (Color online) Calculated total (black) and partial (red) C $2s$ and C $2p$ densities of states (solid, dotted, and dashed lines, respectively) of the crystalline C_{40} allotropes: hyperdiamond (1), hyperlonsdaleite (2), and autointercalated hyperdiamond (3). The energy is given relative to the Fermi energy $E_F = 0$ eV.

$$I(s) = \sum_i \sum_j f_i f_j \frac{\sin(sr_{ij})}{sr_{ij}},$$

where r_{ij} is the distance between i th and j th atoms and f_i and f_j are the atomic scattering factors of the i th and j th atoms. s is the x-ray scattering vector and is equal to $s = 4\pi \sin \theta / \lambda$, 2θ is diffraction angle, and λ is x-ray wavelength (in our calculations 1.542 \AA like for nickel filtered $Cu K\alpha$ radi-

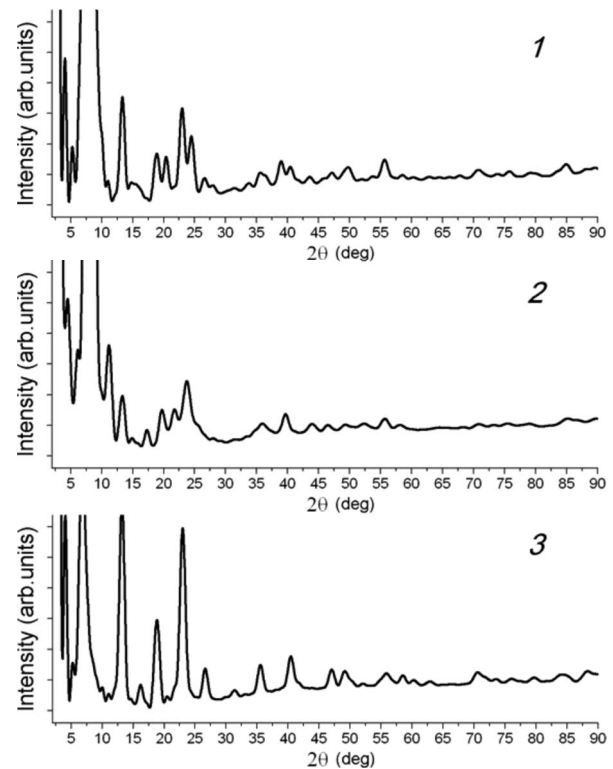


FIG. 4. Theoretical XRD spectra of the crystalline C_{40} allotropes: hyperdiamond (1), hyperlonsdaleite (2), and autointercalated hyperdiamond (3).

tion). The values of the atomic scattering factors were taken from Ref. 21.

Comparing theoretical XRD spectra (Fig. 4) one can see a characteristic feature of XRD for C_{40} AHD: distinct peak at 23.5° , which is similar to (002) peak for graphite and caused by the presence of C_6 hexagons, which belong to the fullerenes of different sublattices and are situated parallel. In XRD spectra of C_{40} HD and HL this pattern is not so much distinct and is separated into a few reflexes.

In conclusion, the atomic models of diamondlike crystalline allotropes composed of C_{40} fullerenes as building blocks (hyperdiamond, hyperlonsdaleite, and so-called autointercalated hyperdiamond) were constructed and their structural, elastic, cohesive, and electronic properties were predicted by means of the DFTB method. We have shown that all these C_{40} allotropes will be energetically stable systems and are

semiconducting. However, their mechanical characteristics differ considerably. Namely, the bulk modulus for the most stable allotrope with higher mass density— C_{40} autointercalated hyperdiamond—is twice larger than bulk moduli for C_{40} hyperdiamond and hyperlonsdaleite.

Besides, special interest represents the bonding between C_{40} molecules inside this allotrope, which has twofold character (covalent and van der Waals), i.e., combines the types of the interactions presented in two main groups of the known fullerites C_n : in molecular crystals ($C_{n \geq 60}$ -based fullerites) and in alternative group of $C_{n < 60}$ -based covalently bonded fullerites. We believe that this finding may stimulate future search for C_n assembled materials with such unconventional “mixed” bonding mechanism.

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